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Current Opinion in Colloid & Interface Science 5 (2000) 56–63

Current Opinion in
Colloid & Interface
Science

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Colloidal crystals as templates for porous materials

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Abstract

Close-packed colloidal crystals are promising precursors for novel materials, but only after appropriate methods are developed to fix their structure. A wide range of advanced materials has recently been synthesized by replicating the structure of colloidal crystals into durable solid matrices. Such materials with structured pores have promise as photonic crystals, catalysts, and membranes, and in a variety of other applications. This paper reviews the methods used in the formation of these materials and likely future trends in the field. © 2000 Elsevier Science Ltd. All rights reserved.

Keywords: Colloidal crystals; Porous materials; Templates

1. Synthesis strategies

The long-range ordering of particles in the structure of colloidal crystals results in a number of unique potentially useful properties, such as optical diffraction and photonic band gaps, maximal packing density and high surface/volume ratio. However, the materials obtained after the particulate arrays are dried are very brittle and can be re-dispersed in water. A fascinating example of how colloidal crystals can be turned into a durable material with remarkable properties is provided by the natural opals, which are formed when the voids between ordered sediments of silica particles are infiltrated by hydrated silica, which then solidifies.

A variety of artificial structures based on colloidal crystals have been synthesized recently using the ‘fixing’ concept suggested by the opals. The primary focus of this paper is to review recent advances in creating novel materials by replicating the structure

of colloidal crystals into durable matrices. In this approach (Fig. 1), the colloidal crystals are assembled in order to serve as templates, the voids of which are infiltrated by material that solidifies there. The original colloidal particles are subsequently removed, leaving behind a new material with pores that preserve the most valuable property of the colloidal crystals — the long-ranged periodic structure. A major advantage of the colloidal crystal template method is the ability to control the dimensions of the pores easily by varying the size of the beads in the templates.

Latex and silica microspheres are the two major types of particles that are usually used for the colloidal crystal assembly, as they can be obtained both highly monodisperse and relatively cheaply. Central to the materials synthesis is the design of procedures for assembling the crystals and subsequently infusing them with appropriate media. A summary of the crystallization methods used by different investigators is represented in Fig. 2. The simplest method is gravitational sedimentation of the particles, usually combined with drying of the suspension from above. The process can be accelerated and the quality of the materials can be improved via centrifugation (Fig. 2b).

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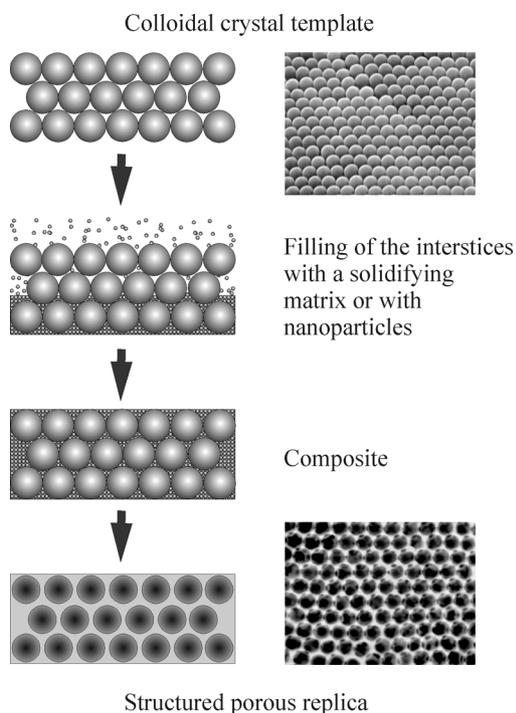


Fig. 1. Schematic of the general procedure for replicating the structure of colloidal crystals into porous materials.

Another way to speed the process up is by filtration (Fig. 2c), which also allows easy washing and subsequent infusion with different media. Of certain interest is the formation of crystalline sheets of specific thickness, for which the convective assembly method [2–4] (Fig. 2d) has been used after recent modifica-

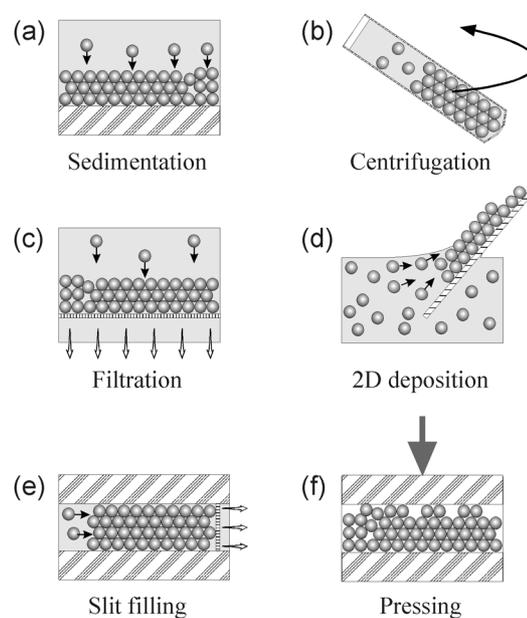


Fig. 2. Schematics of methods used for assembling the colloidal crystalline templates.

tions and improvements [5]. Colloidal crystals can also form when the particles are confined in the thin film between two solid boundaries [1]. This approach has also recently been used to assemble closely packed crystals, by filtering suspensions into a thin slit between two solid plates (Fig. 2e) [6,7]. Finally, ordered structures between solid surfaces can be formed by pressing and compaction of particles in the dry state (Fig. 2f).

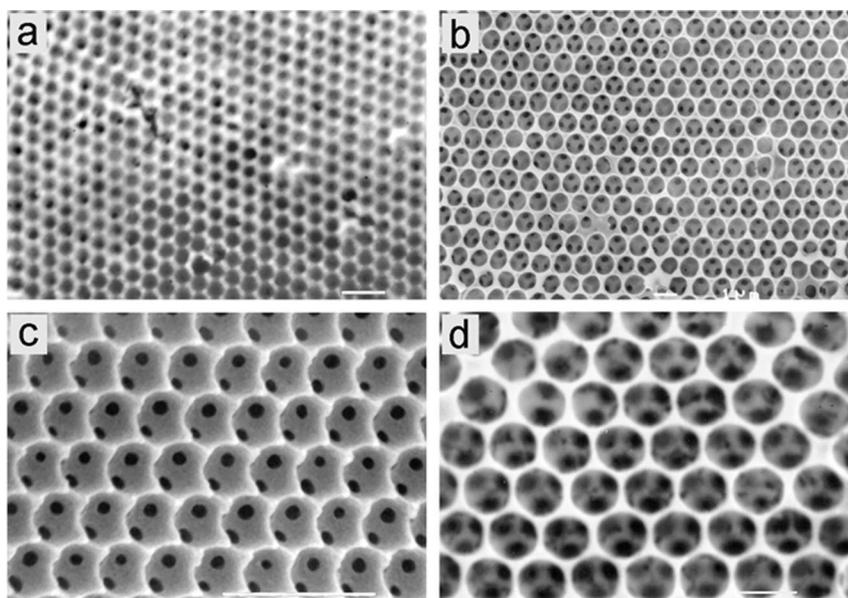


Fig. 3. Scanning electron micrographs illustrating the similarity in the structure of materials formed from different precursors. (a) Porous silica sample, reprinted with permission from Velev et al. [11]; (b) Porous polymer membrane, photo courtesy of Gates et al. [30]; (c) Inverse opal from phenolic resin, photo courtesy of Zakhidov et al. [33]; (d) Sample of metallic gold similar to those in Velev et al. [35]. Scale bars = 1 μm .

In all of the above methods, the transition to the ordered state is induced by the increased concentration of the particles in the vicinity of a flat surface. The phases formed are usually built up of polycrystalline domains in different orientations, for several reasons: (i) nucleation tends to occur concurrently at many locations; (ii) the high friction within the arrays inhibits re-arrangement; and (iii) impurities are present that distort the lattice. The typical lattice types in these domains are randomly-stacked hexagonal close-packed planes (r.h.c.p.), and face-centered cubic (f.c.c.) [8,9]. Square arrays, which may correspond to body centered cubic (b.c.c.) packing, are formed occasionally in the interstices between the growing hexagonal domains, or in films of thickness commensurate with a discrete number of squarely packed layers [1]. The growth of defectless crystalline templates with desired symmetry and orientation is one of the major challenges in the field. One way to achieve this may be the use of micropatterned surfaces to promote epitaxial-like growth [10•].

The solidified structure in the pores between the microspheres can be formed by, for example, polymerization [11•,12•] or sol-gel hydrolysis [13•,14•,15•] of a liquid precursor. More recently, structures have been grown via electroless or electrochemical deposition [16•,17•] or precipitation [18•]. A conceptual extension of the method that avoids complex chemistry is to fill the interstices of the colloidal template with smaller colloidal particles [19•], which also leads to a structure with a hierarchical porosity on both mesoscopic and macroscopic scales. The size of the large pores can be manipulated via the diameter of the templating particles, while the size of the small pores and the overall specific surface area are determined by the size of the small particles.

In the final step of the process, the colloidal crystal templates are removed from the composite material either by calcination or by chemical or physical dissolution. The pores left in place of the particles are arranged in ordered three-dimensional arrays that represent a negative replica of the original colloidal crystal (Fig. 1). The morphology of the structures is broadly similar irrespective of the type of solid matrix used (Fig. 3). The various materials synthesized to date are reviewed below, grouped by their chemical composition and hence potential function.

2. Materials based on inorganic oxides

The first demonstration of replicating the ordered structure of colloidal crystals into a stable matrix yielded structured porous silica [11•,20]. The templates were assembled by filtering diluted suspensions of latex microspheres through smooth membranes.

The surfaces of the microspheres were functionalized by adsorption of cationic surfactant from solution, which served to initiate the polymerization of $\text{Si}(\text{OH})_4$ to solid silica after the crystal was subsequently infused with aqueous silica solution [20]. The latex particles were removed from the composite via calcination. The products of the mineralization are porous low-density silica flakes (Fig. 3a). The range of pore sizes obtained by varying the size of the latexes used, 150 nm to 1 μm , is typical for such materials.

In a separate leading study, Imhof and Pine [13•] described how colloidal crystal-like assemblies of densely packed monodisperse non-aqueous emulsion droplets can serve as templates for the formation of porous titania, zirconia and silica. The pores are less ordered and uniform than with latex microspheres, but the method has technological potential due to the use of simple easily obtained templates. In addition, the sol-gel alkoxide hydrolysis method employed for depositing the oxide matrix had later proved useful with latex particle templates. The formation of macroporous materials from simple but generally disordered templates has also been reported by Davis et al. [21] and Antonietti et al. [14•].

The versatility of the principle has been demonstrated both in the range of materials made and in the ability to design specific functional and structural characteristics into them. Holland et al. [15•] assembled latex crystals by both filtration and centrifugation (Fig. 2a,b) and used them as templates for porous titania, alumina and zirconia via the sol-gel technique. This group has adapted the method further to form structures of a wide variety of chemical compositions, including oxides of W, Fe, Sb, Zr/Y, aluminophosphates, silicates, carbonates, zeolites and others [18•,22•,23]. Doping of porous titania with metals such as Co has also been reported [24•].

The first study specifically to target the formation of a material with photonic crystal properties, dense porous titania, was that of Wijnhoven and Vos [25•]. The templates are assembled via centrifugation, and the titania structure is formed by the sol-gel method. The high refractive index of the titania is a prerequisite for a remarkably wide reflectance peak of the calcined structure, although the samples obtained do not display a full photonic band gap.

Greater structural complexity has been attained by complementing the colloidal crystal templates with other templating techniques to create hierarchically ordered porous oxides [26•]. The smallest ordered mesopores (~ 10 nm) are formed by templating by surfactant block copolymers analogously to the MCM materials templated by surfactant assemblies. The mid-ranged ordered pores (~ 100 nm) are templated by the arrays of latex microspheres in the colloidal

crystals. The largest features ($\sim 1 \mu\text{m}$) are created by molding with a PDMS master.

Porous silica and titania have also been assembled from nanoparticles. In the direct methods formulated by Subramania et al. [27] and Subramanian et al. [28], the latex particles are mixed with TiO_2 or SiO_2 nanoparticles. The latex is crystallized via drying and sedimentation, during which process the inorganic nanoparticles fill the interstices in the crystal. The simplicity of this method and its potential for scaling up make it attractive for practical applications.

3. Structured porous polymers

The use of colloidal crystals as templates for forming porous polyurethane membranes was first demonstrated by Park and Xia [12•,29]. The latex templates are assembled by injection and accumulation between solid plates (Fig. 2e). The polyurethane precursor is UV-polymerized and the latexes are removed by selective dissolution in toluene, leaving structures analogous to those in Fig. 3b. These authors later extended the fabrication procedure to silica spheres [30•]. Johnson et al. [31•] synthesized porous polymers using microspheres $< 100 \text{ nm}$ in diameter, closing the gap of pore sizes attainable via colloidal crystal templating and the conventional mesoporous MCM materials. The templates in this case were assembled via pressing in the solid state (Fig. 2f). Finally, Jiang et al. [32] modified the convective assembly method (Fig. 2d) to prepare free-standing, optically active porous films from a wide variety of polymers.

4. Materials from carbon and semiconductors

Zakhidov et al. [33•] have developed procedures for using crystalline templates to form a family of structured porous carbons with remarkable optical properties. The templates are artificial opals assembled from SiO_2 microspheres, which are embedded into carbon phases by three alternative routes: (i) by infiltrating the crystals with phenolic resin, removing the microspheres, and pyrolyzing the resin structure to glassy carbon; (ii) by chemical vapor deposition (CVD) of graphitic carbon; and (iii) by diamond-seeded CVD from plasma. The carbon ‘inverse opals’ obtained are highly conductive, show intense opalescence from the ordered arrays of holes and may have a photonic band gap in the infrared region.

Structured porous semiconductors are of significant interest because of the quantum ‘dots’ and ‘wells’ they display and because their high refractive index holds promise for use in photonic crystals. Vlasov et

al. [19•] were the first to assemble a secondary structure from nanoparticles and prepare a porous material from semiconductor CdSe quantum dots. The effective refractive index of the material assembled is, however, lower than that of solid semiconductors due to the presence of air in the pores between the nanocrystals. The problem of creating an optically dense, continuous, semiconductor structure is solved in the electrochemical method reported by Braun and Wiltzius [17•]. Here, the template crystals of latex or silica microspheres are assembled via sedimentation onto conductive indium tin oxide (ITO) electrodes, and then CdSe and CdS are grown electrochemically in their interstices via electrodeposition. Structures made by this approach should display deep photonic band gaps and good mechanical stability.

5. Structured porous metals

Due to the importance of metals in technological applications, structured porous metals may find application in a variety of areas, particularly electronics and optoelectronics. The direct infusion of a heat-resistant crystal with a molten metal presents technical problems due to the high temperatures and pressures required. Although such methods may be reported in the future, the methods published to date create the metallic structure via alternative procedures. Yan et al. [34•] assembled latex crystals by centrifugation and impregnated the template with a nickel salt that was then oxidized to NiO and finally reduced to metallic Ni in a hydrogen atmosphere. The highly porous nickel obtained can be useful in catalysis and in electrochemical electrodes.

Using a totally ‘wet’ approach, Jiang et al. [16•] synthesized a variety of porous metals by electroless deposition. Silica colloidal crystals were assembled in thin wetting films (Fig. 2d) and functionalized with gold nanocrystals that provide the nucleation sites for electroless deposition. The method has been used to form samples from Ni, Cu, Ag, Au and Pt, which have remarkable optical properties due to the long-ranged ordered porous structure on the surface.

Our wet method [35•] produces a templated gold structure assembled entirely from a suspension of gold nanoparticles. This is made possible by assembling the latex crystal by filtration through a membrane of pore size small enough to retain both the latex as well as the gold particles subsequently added, while still allowing a reasonably high flux of water. Thus, a mesoscopically porous gold structure is grown in the interstices of the latex crystals. Two alternative procedures are used to remove the latex beads from the composite, yielding porous metal with different properties. One procedure is calcination at 300°C ,

which leads to some melting and fusion of the nanoparticles, so that only the larger pores templated by the polymer beads are preserved. Alternatively, a hierarchical mesoporous–macroporous structure is created when the latex templates are removed by chemical oxidation or solvent dissolution at room temperature. Both of these metallic materials exhibit

brightly colored reflections in incident illumination and are highly conductive. Examples of the discrete morphology of these samples and their hierarchical porosity are shown in Fig. 4.

6. Current challenges and future directions

In just a few years the fabrication of porous materials using colloidal crystal templates has become a rapidly growing area in nanomaterials. As shown here, nearly all classes of organic and inorganic materials and metals have already been templated into porous ordered structures. The design and synthesis of these structures is a fascinating, intellectually challenging problem, but the interest in these materials is also rooted in their wide array of potentially usable applications.

First, the method has produced materials with closely spaced pores of sizes from < 100 to > 1000 nm, a range that cannot be obtained by the earlier techniques such as templating via surfactant micelles and liquid crystals. The uniformity and the interconnection of the densely packed pores make some of these materials usable as efficient filtration membranes [30•]. The three-dimensional materials of hierarchical porosity have potential applications in advanced catalysis, where the larger pores allow rapid mass transport, while the small ones provide high surface area. Promising directions are the synthesis of silicates with zeolitic microporous frameworks [23] and doping and modification of the bulk and surfaces of the pore walls [22•,24•]. Theoretical research on the dynamics of formation of such materials is under way [36]. Interesting catalytic and other applications can be foreseen by modifying the method to form membranes and layers of alternating chemical composition and by implanting catalytic particles within the pores. An additional level of structural complexity can be obtained via subsequent disassembly [37] or reverse templating [38].

Second, the long-ranged ordering of the submicrometer pores opens a wide variety of potential applications in areas such as optical information processing and storage, advanced coatings and emerging nanotechnologies. The most ‘visible’ of these applications is as photonic crystals. Such structures, with three-dimensional periodicity on a length scale comparable to that of light, could be used to make microscopic lasers and efficient light emitting diodes, and be used as miniature waveguides or mirrors in optical processing devices [39–42]. The alternative micromachining methods for creating photonic structures are expensive, and the formation of similar materials via self-assembly and templating is a lucrative undertaking. It has been shown that the structured porous materials

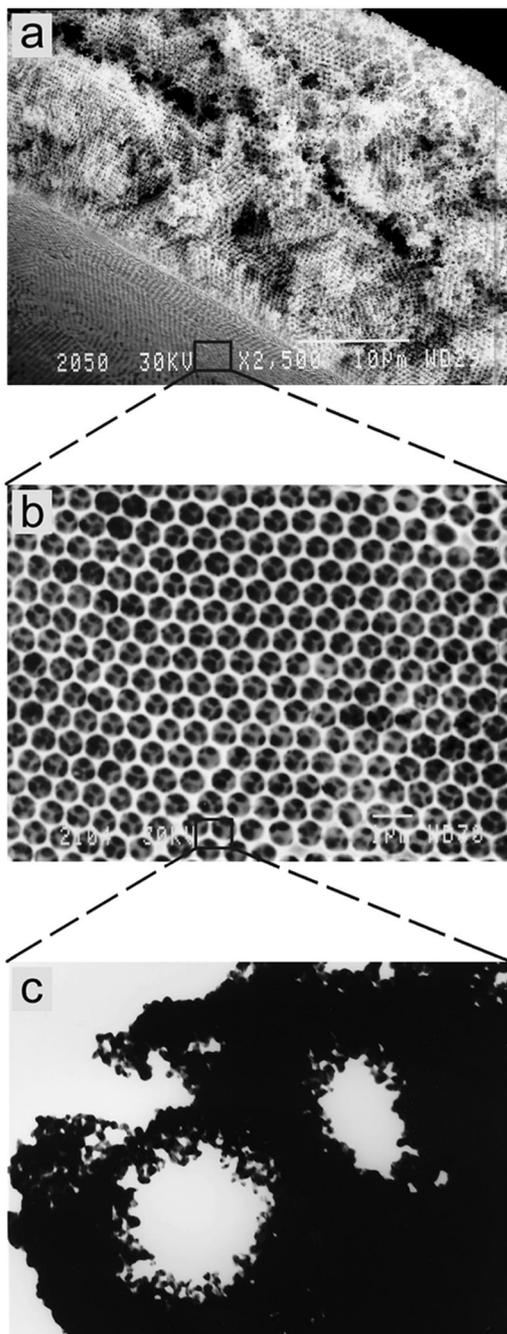


Fig. 4. Electron micrographs at different magnifications, illustrating the hierarchical structure of the meso-macroporous material assembled from metallic nanoparticles. (a) Low magnification SEM across the edge of a metallic flake; (b) A representative area on the surface; (c) TEM demonstrating that the structure is assembled from nanoparticles. Adapted with permission from Velev et al. [35].

discussed here can possess the desired full photonic bandgap when created from a matrix with high refractive index [17•,19•,42]. The synthesis of such structures is still a challenge, as few materials have such a high refractive index and it is necessary to grow these in a continuous manner [17•]. Another obstacle that has been solved in principle [10•], but still remains a technological challenge, is the growth of defectless colloidal crystals of desired symmetry and orientation.

A major direction that warrants exploration is the use of colloidal crystals as templates in fabricating supported thin films and for surface modification of materials. One topic that has been relatively well studied previously is the use of thin colloidal crystals as non-lithographic masks for forming surface patterns via deposition of metal through the layers [43–47]. Two-dimensional colloidal templates have recently been used in preparing porous TiO₂ surfaces with enhanced photocatalytic activity [48•]. Thin semitransparent films from porous metals may have interesting transmission properties arising from the surface plasmons in the metallic layers [49•,50,51]. Recently, we have been able to modify the method for preparing porous metals so as to form two-dimensional structured metallic layers only 1–3 pores thick deposited on a glass substrate, which are semitransparent and show colors by diffraction [52]. One promising application of such structured porous metallic films is as substrates for surface enhanced Raman spectroscopy (SERS) [53–55], where the cheap and simple assembly method has significant advantages over the microfabrication techniques used at present [55]. Given the interest and the strongly competitive research in this field, exciting new ideas and practical applications are likely to emerge rapidly in the future.

Acknowledgements

Our research referred to here has been performed with the essential collaboration of Peter Tessier, Raul Lobo and Eric Kaler. The partial financial support of the Department of Chemical Engineering, University of Delaware, is acknowledged.

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